

Electrodeposition and characterization of Cu-TiO₂ nanocomposite coatings

S. Ramalingam · V. S. Muralidharan · A. Subramania

Received: 17 December 2008 / Revised: 12 May 2009 / Accepted: 18 May 2009 / Published online: 19 June 2009
© Springer-Verlag 2009

Abstract Cu-TiO₂ nanocomposites were prepared by electrodeposition method onto copper substrate using an acid copper plating bath containing dispersed nanosized TiO₂. The composition of codeposited TiO₂ nanoparticles in the composite coatings was controlled by the addition of different concentrations of TiO₂ nanoparticles in the bath solution. The average crystallite size was calculated by using X-ray diffraction analysis and it was ~32 nm for electrodeposited copper and ~33 nm for Cu-TiO₂ composite coatings. The crystallite structure was fcc for electrodeposited copper and Cu-TiO₂ nanocomposite coatings. The surface morphology and composition of the nanocomposites were examined by scanning electron microscopy and energy dispersive X-ray spectroscopy analysis. The effect of TiO₂ content on the corrosion and wear resistance properties of the nanocomposite coatings was also presented. The codeposited TiO₂ nanoparticles in the deposit increased the corrosion and wear resistance, which were closely related with TiO₂ content in the nanocomposites. The wear resistance and microhardness of the Cu-TiO₂ nanocomposite coatings were higher than electrodeposited copper. The corrosion resistance property of the electrodeposited copper and Cu-TiO₂ nanocomposite coatings was evaluated by electrochemical impedance and Tafel polarization studies. Cu-TiO₂ composite coatings were more corrosion resistant than electrodeposited copper.

Keywords Nanocomposites · Electrodeposition of copper · Cu-TiO₂ nanocoating · Composite coatings · Corrosion resistance

Introduction

Nanocomposite coatings refer to electrolysis in which nanosized particles are suspended in an electrolyte and are embedded in the electroformed solid phase, imparting special properties depending on the degree and type of nanoparticle incorporation in the deposit. Electrodeposited composite coatings exhibit enhanced materials properties such as microhardness, wear resistance, and corrosion resistance than the electrodeposited metal coatings [1, 2]. These enhanced properties depend mainly on the nature of the codeposited particles, as well as the distribution of the particles in the metallic matrix [3]. Electrodeposition technique is one of the methods used to fabricate nanocomposite coatings. Many literatures have demonstrated electrodeposited copper- or copper alloy-based nanocomposite materials [4–10]. TiO₂ has many potential applications in various industrial and domestic fields including metallic and organic coatings, catalysis as catalyst support, photovoltaic cells, and wastewater treatment. TiO₂ has also been used to reinforce metallic coatings and it improves wear resistance, hardness, and other properties such as corrosion resistance [11, 12]. TiO₂ codeposition was carried out with Ni, Zn, Ag, and Cu as the metallic components [13–20]. The interest in copper composite coatings has been increased for their potential engineering applications resulting from the outstanding properties of wear resistance, anticorrosion, and self-lubrication. To achieve these properties, various inert particles such as SiC, SiO₂, graphite, Al₂O₃, MoS₂, TiO₂, PTFE, etc. are tried so far [4, 21–27].

S. Ramalingam · A. Subramania (✉)
Advanced Materials Research Laboratory,
Department of Industrial Chemistry, Alagappa University,
Karaikudi 630 003, India
e-mail: a_subramania@yahoo.co.in

V. S. Muralidharan
Alagappa Chettiar College of Engineering and Technology,
Karaikudi 630 004, India

Celis et al. reported the kinetics and mechanism of copper-Al₂O₃ nanocomposite coatings [25, 28, 29]. Lozano-Morales et al. studied the effect of Al₂O₃ nanopowder on copper electrodeposition [30]. Benea et al. reported the corrosion behavior of copper and copper matrix composite coatings, with ZrO₂ particles embedded by electrodeposition from an acid copper sulfate plating bath [31]. Yang et al. reported the preparation of single-walled carbon nanotube-reinforced copper composite coatings by electrochemical deposition method [32]. Guo et al. reported the pulse plating of copper-ZrB₂ composite coatings [33].

In this present work, a novel nanostructure metallic composite coatings consisting of nanocrystalline copper matrix (average crystallite size=32 nm) and dispersed titania nanoparticles (average crystallite size=25 nm) have been electrodeposited onto copper substrates from an acid copper sulfate bath. The crystallite size and structure of the electrodeposits were calculated by using X-ray diffraction (XRD) analysis. The surface morphology and composition of copper and Cu-TiO₂ nanocomposite coatings were examined by scanning electron microscopy (SEM) and energy dispersive X-ray spectroscopy (EDAX) analysis. The microhardness and the wear resistance of Cu-TiO₂ nanocomposite coatings were measured by Vickers microhardness tester and Taber abrader tester, respectively. Finally, the corrosion resistance behavior of electrodeposited copper and Cu-TiO₂ nanocomposite coatings was evaluated by electrochemical impedance and Tafel polarization studies.

Experimental methods

Electrodeposition of Cu-TiO₂ nanocomposites

Cu-TiO₂ nanocomposite coatings were electrodeposited from a suspension of TiO₂ nanoparticles in a copper sulfate bath. The optimized bath composition for Cu-TiO₂ composite plating bath was copper sulfate (0.3 M) and sulfuric acid (1.3 M). The TiO₂ nanoparticles (25 nm) were added to the plating bath solution from 25 to 100 g/L with continuous agitation for 4–6 h to ensure uniform dispersion of the nanoparticles in the plating bath [34]. Electrocomposite coatings were carried out using a copper bar of high purity as the anode and polished copper plates of 4.0 cm² area as the cathode materials for composite plating. Gelatin (1.0 g/L) was used as an additive in the bath solution. The electroplating parameters such as the current density and pH of the bath were fixed as 5.0 A/dm² and 0.2, respectively, at 30 °C. Electrodeposited copper was also obtained under the same condition from an electrolyte without TiO₂ nanoparticles.

Characterizations of Cu-TiO₂ nanocomposites

The crystallite size and structure of the electrodeposited copper and Cu-TiO₂ nanocomposite coatings were measured by XRD analysis. Surface morphology of electrodeposited copper and Cu-TiO₂ nanocomposite coatings was examined by SEM. The composition of the Cu-TiO₂ nanocomposite coatings were measured by EDAX analysis.

The microhardness of the mechanically polished electrodeposited copper and Cu-TiO₂ nanocomposite coatings was measured by Vickers microhardness tester using a load of 50 g for the duration of 15 s. The wear test was conducted on a sliding wear machine of Taber abrader tester with a rotation rate of 1,000 cycles under an applied load of 1 kg.

The corrosion resistance behavior of the electrodeposited copper and Cu-TiO₂ nanocomposite coatings was evaluated by electrochemical impedance spectroscopy. The coated specimen of 1.0 cm² area was used as working electrode, the standard calomel electrode (SCE) was used as reference electrode, and the Pt foil acted as a counter electrode. These three electrodes were immersed in 3.5% NaCl solution in a three-electrode cell assembly. The time interval of 10 min was given to attain a steady-state open-circuit potential (OCP) and then the impedance measurements were carried out at the OCP. A sine wave voltage of 10 mV was superimposed on the rest potential. The real part (*Z'*) and the imaginary part (*Z''*) were measured in the frequency range of 100 kHz to 10 mHz. From the Nyquist plots, the value of charge transfer resistance (*R*_{ct}) and double layer capacitance (*C*_{dl}) were measured.

The corrosion resistance behavior of the electrodeposited copper and Cu-TiO₂ nanocomposite coatings of 1.0 cm² area was evaluated by Tafel polarization studies. The cell configuration used for this study was same as that used in the impedance measurements. The coated specimen of 1.0 cm² area was immersed in 3.5% NaCl solution for the duration of 10 min to attain a steady-state potential. The electrode was then polarized cathodic to anodic direction in the potential range of *E*_{corr}±250 mV from the OCP at a scan rate of 1 mV/s. The corrosion current densities (*i*_{corr}) were determined by extrapolating the linear portion of the anodic and cathodic polarization curves. The polarization curves were recorded at least twice to prove the reproducibility of the results. The corrosion rate (CR) was calculated from the corrosion current by using the following equation [35]:

$$CR = k \left(\frac{i_{\text{corr}}}{\rho} \right) EW$$

where CR is given in millimeters per year, $k=3.27 \times 10^{-3}$ mm g/μA cm year, $\rho=8.96$ g/cm³ is the density of Cu, and EW=31.77 is the equivalent weight of Cu.

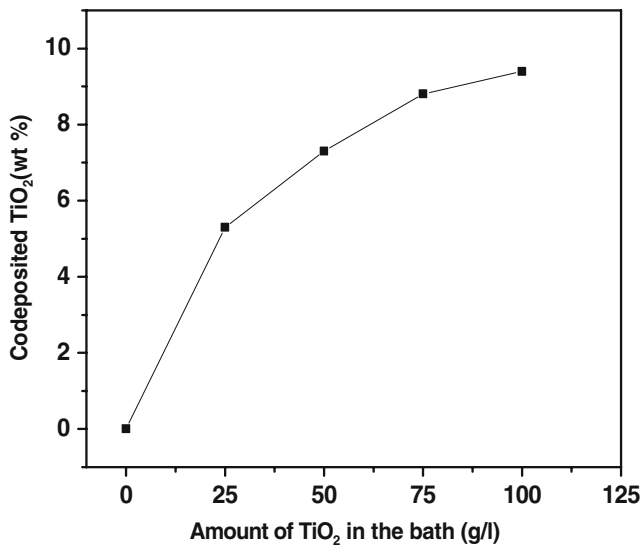


Fig. 1 Effect of amount of TiO₂ in the bath (in grams per liter) on the weight percentage of TiO₂ in the nanocomposite coatings

Results and discussion

Effect of particle concentration

The relationship between the amount of codeposited TiO₂ and the concentration of TiO₂ nanoparticles in the plating bath at the fixed current density of 5.0 A/dm² at 30 °C is shown in Fig. 1. The amount of codeposited TiO₂ nanoparticles in the composite coating increased with TiO₂ concentration in the plating bath [2, 3]. The maximum amount of codeposited TiO₂ nanoparticles was achieved at

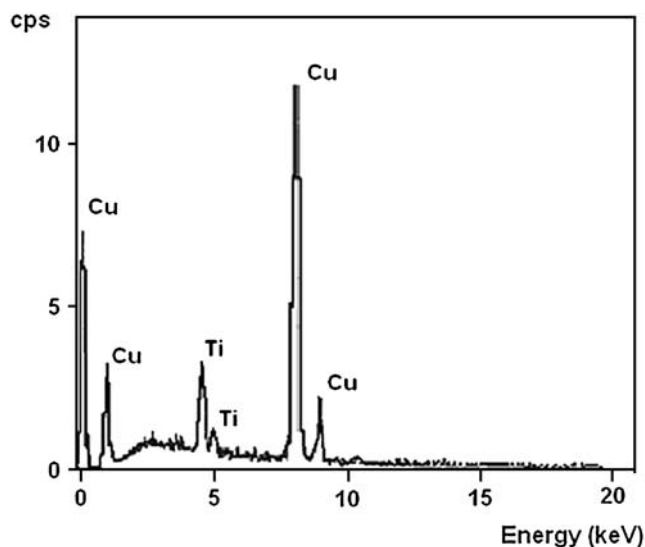


Fig. 2 EDAX analysis spectrum of electrodeposited Cu-TiO₂ nanocomposite coating

Table 1 EDAX analysis of Cu-TiO₂ (7.3 wt.%) nanocomposite coating obtained from 50 g/L of TiO₂ in the bath

Element	Element %			Average %
	First spot	Second spot	Third spot	
Ti (as TiO ₂)	7.30	7.32	7.34	7.32
Cu	92.70	92.68	92.66	92.68

the concentration of 100 g/L of TiO₂. This was confirmed by EDAX compositional analysis. The curve is quite similar to the well-known Langmuir adsorption isotherms, supporting a mechanism based on an adsorption effect. The codeposition of TiO₂ nanoparticles on the cathode surface was suggested by Guglielmi’s two-step adsorption model [36, 37]. Once the particles are adsorbed, metal begins building around the cathode slowly, encapsulating, and incorporating the particles. The highest concentration of TiO₂ nanoparticles on the codeposit was due to saturation in adsorption on the cathode surface.

EDAX analysis

A representative EDAX diagram for the Cu-TiO₂ nanocomposite coating obtained by the addition of 50 g/L of TiO₂ nanoparticles is shown in Fig. 2. The EDAX analysis of Cu-TiO₂ nanocomposite coatings gives the average elemental percentage of Ti and Cu as 7.32% and 92.68%, respectively. It confirms the presence of codeposited TiO₂ nanoparticles on the copper matrix (Table 1).

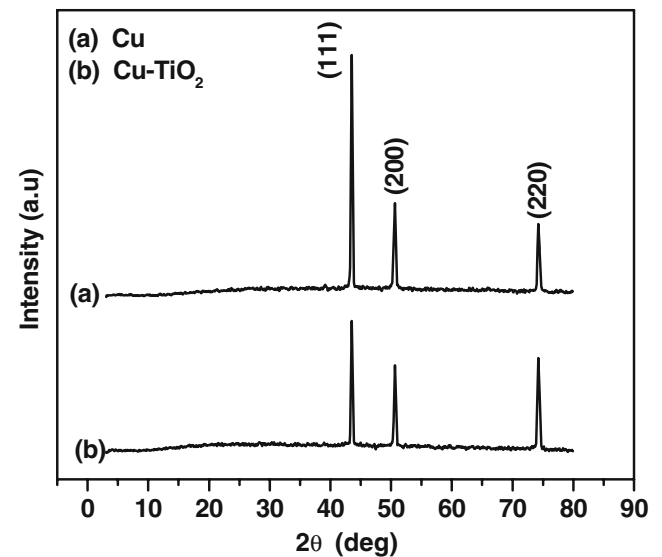


Fig. 3 XRD patterns of both electrodeposited copper and Cu-TiO₂ nanocomposite coatings. **a** Electrodeposited copper, **b** Cu-TiO₂ nanocomposite coating

Table 2 XRD parameters for electrodeposited copper and Cu-TiO₂ nanocomposite coatings

Electro deposits	<i>d</i> -spacing (Å°)		Miller indices (<i>h k l</i>)	Lattice parameter (a)		Str./phase	Average crystallite size (nm)
	Observed	Standard		Observed	Standard		
Copper	2.0794	2.0883	(111)	3.605	3.615	fcc	32
	1.8020	1.8082	(200)				
	1.2763	1.2780	(220)				
Cu-TiO ₂ (7.3 wt.%) composite	2.0794	–	(111)	3.605	3.615	fcc	33
	1.8024	–	(200)				
	1.2761	–	(220)				

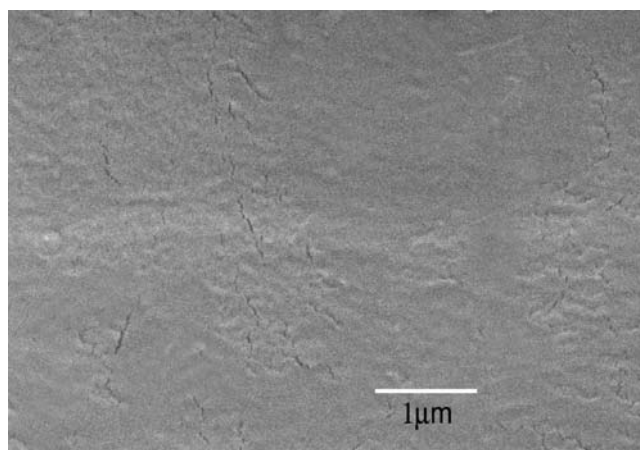
X-ray diffraction analysis

The crystallite size and structure of the electrodeposited copper and Cu-TiO₂ nanocomposite coatings were calculated by using XRD analysis (Fig. 3). Its corresponding

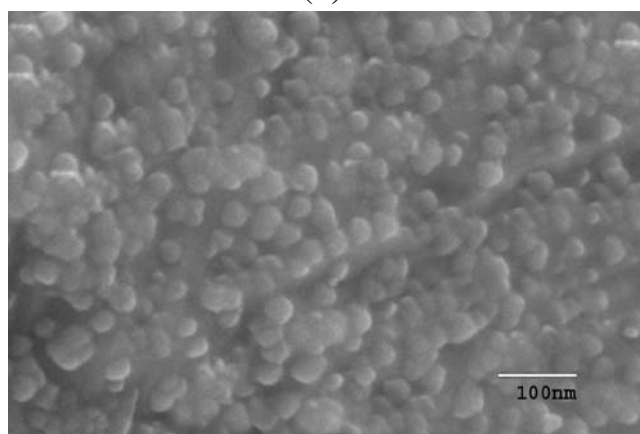
XRD data is given in Table 2. The average crystallite size calculated from the XRD pattern using the Scherrer equation was ~32 nm for copper and ~33 nm for Cu-TiO₂ nanocomposite electrodeposits. The structure of electrodeposited copper and Cu-TiO₂ nanocomposite coatings were crystalline fcc which was confirmed from the JCPDS standards [38], but only noticeable differences in the intensity of (111), (200), and (220) peaks were observed for the Cu-TiO₂ nanocomposite coating compared to electrodeposited copper. The effect of the inclusion of TiO₂ nanoparticles in the copper matrix had only negligible influence on the internal stress, which is evident from the XRD results.

SEM studies

SEM micrographs of the electrodeposited copper and Cu-TiO₂ nanocomposite coatings are shown as Fig. 4. Both the coatings were electrodeposited at a current density of 5.0 A/dm² and layer thickness of about



(a)



(b)

Fig. 4 SEM photographs of electrodeposited copper and Cu-TiO₂ nanocomposite coatings. **a** Electrodeposited copper, **b** Cu-TiO₂ nanocomposite coating

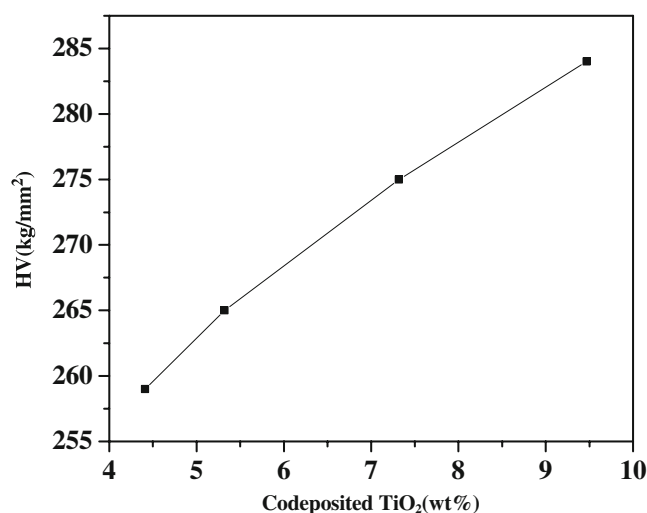


Fig. 5 Effect of the amount of codeposited TiO₂ on the microhardness of Cu-TiO₂ nanocomposite coatings

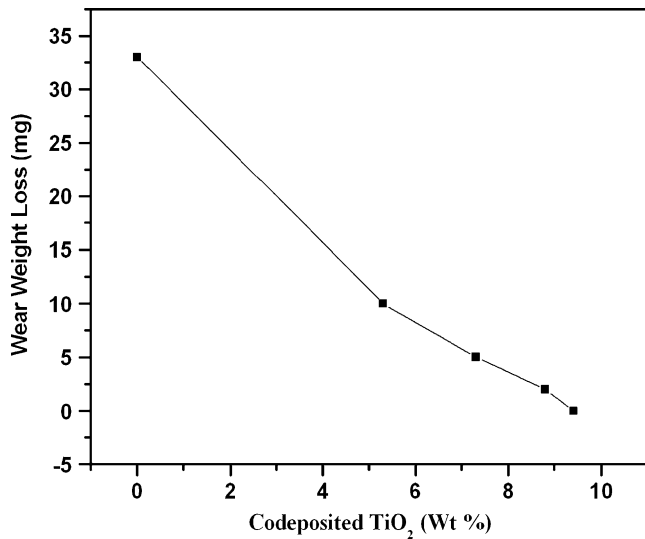


Fig. 6 Effect of the amount of codeposited TiO₂ on the wear weight loss of Cu-TiO₂ nanocomposite coatings

60 μm. The electrodeposited copper showed regular surface and fine crystallites as observed in Fig. 4a and the Cu-TiO₂ nanocomposite coating was more compact than the electrodeposited copper and consisted of smaller and spherical-sized grains as observed in Fig. 4b. It was evident that the TiO₂ nanoparticles were uniformly distributed in the copper matrix by electrodeposition, which was consistent with the EDAX results (Fig. 2). Cu-TiO₂ nanocomposite coatings possessed enhanced mechanical as well as corrosion resistance properties with TiO₂ nanoparticle in the deposit. This is because of the uniform distribution of TiO₂ in the deposit.

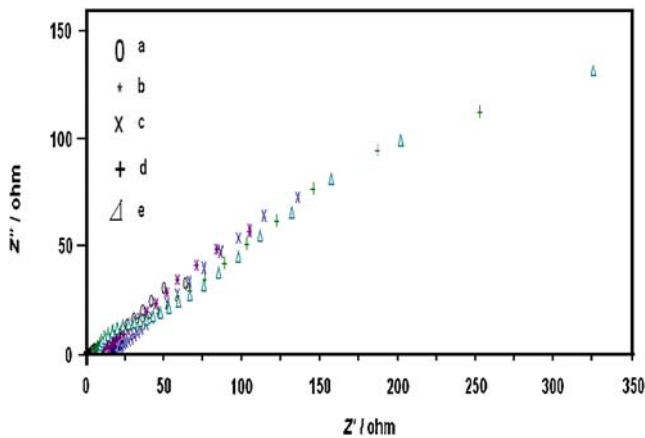


Fig. 7 Impedance plots of electrodeposited copper and Cu-TiO₂ nanocomposite coatings. *a* Electrodeposited copper, *b* Cu-TiO₂ (5.3 wt.%), *c* Cu-TiO₂ (7.3 wt.%), *d* Cu-TiO₂ (8.8 wt.%), *e* Cu-TiO₂ (9.4 wt.%)

Table 3 Impedance parameters for electrodeposited copper and Cu-TiO₂ nanocomposite coatings

Material	TiO ₂ (wt.%)	R _{ct} (ohm/cm ²)	C _{dl} (μF/cm ²)
Electrodeposited copper	0	220	72.37
Cu-TiO ₂ nanocomposite coatings	5.3	660	24.12
	7.3	850	18.73
	8.8	1,000	15.92
	9.4	1,150	13.84

Microhardness and wear resistance

Vickers microhardness measurements were performed for electrodeposited copper and Cu-TiO₂ nanocomposite coatings as shown in Fig. 5. Microhardness of the Cu-TiO₂ nanocomposite coating was 25 kg/mm² higher than that of electrodeposited copper. The microhardness of the nanocomposite coatings increased with TiO₂ particles content in the copper matrix. The enhancement in the hardness of Cu-TiO₂ nanocomposite coatings was due to the strengthening effect caused by the dispersion of TiO₂ nanoparticles in the composite coatings, which impeded the motion of dislocation in the metallic matrix. The Cu-TiO₂ nanocomposite coating exhibited higher hardness values at increased content of the codeposited TiO₂ particles. The improved friction and wear properties were due to the hardened matrix.

The wear resistance of the electrodeposited copper and Cu-TiO₂ nanocomposite coatings were measured by using

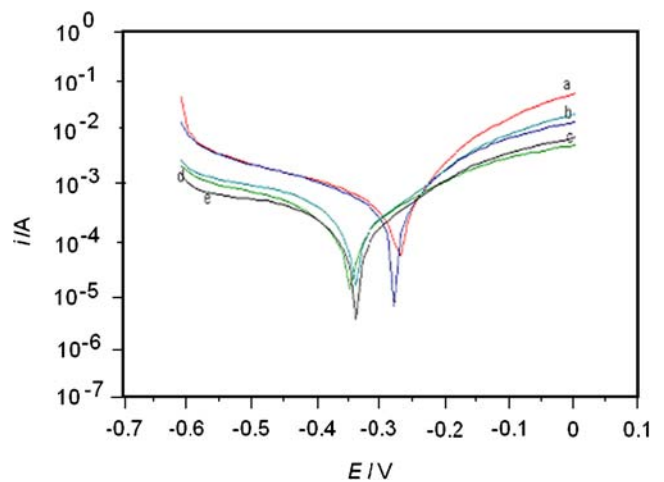


Fig. 8 Tafel polarization plots of electrodeposited copper and Cu-TiO₂ nanocomposite coatings. *a* Electrodeposited copper, *b* Cu-TiO₂ (5.3 wt.%), *c* Cu-TiO₂ (7.3 wt.%), *d* Cu-TiO₂ (8.8 wt.%), *e* Cu-TiO₂ (9.4 wt.%)

Table 4 Tafel polarization parameters for electrodeposited copper and Cu-TiO₂ nanocomposite coatings

Material	TiO ₂ (wt.%)	E_{corr} (V) vs SCE	i_{corr} ($\mu\text{A}/\text{cm}^2$)	CR (mm/year)
Electrodeposited copper	0	-0.271	101.6	1.17
Cu-TiO ₂ nanocomposite coatings	5.3	-0.286	68.25	0.80
	7.3	-0.343	41.01	0.47
	8.8	-0.338	34.38	0.40
	9.4	-0.339	6.83	0.08

the Taber abrader tester as shown in Fig. 6. From this result, it was found that the Cu-TiO₂ nanocomposite coatings exhibited more wear resistance than electrodeposited copper coating.

Corrosion resistance measurements

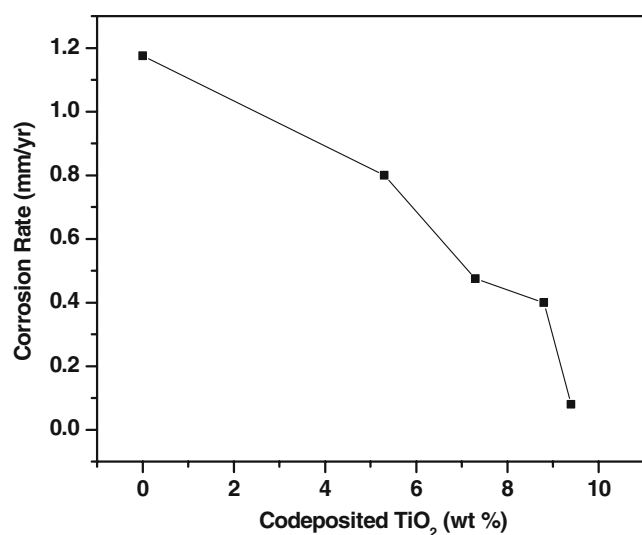
Electrochemical impedance studies

Nyquist plots were obtained for the electrodeposited copper and Cu-TiO₂ nanocomposite coatings to get the value of charge transfer resistance (R_{ct}) and double layer capacitance (C_{dl}) as shown in Fig. 7. The charge transfer resistance (R_{ct}) values for Cu-TiO₂ nanocomposite coatings increased and the double layer capacitance (C_{dl}) values decreased with TiO₂ in the composite coatings (Table 3). It revealed that the Cu-TiO₂ nanocomposite coating was more corrosion resistant than electrodeposited copper coating.

Polarization measurements

The corrosion potentials (E_{corr}), the corrosion current (i_{corr}), CR, and the Tafel slopes b_a and b_c for copper and Cu-TiO₂ nanocomposite coatings were calculated from the Tafel polarization curves (Fig. 8) and are given in Table 4. It can be seen from the table that the corrosion current (i_{corr}) decreased in all the Cu-TiO₂ nanocomposite coatings compared to electrodeposited copper. The corrosion potential in the case of Cu-TiO₂ nanocomposites had shown a negative shift, confirming the cathodic protective nature of the coatings.

The CR of electrodeposited copper and Cu-TiO₂ nanocomposite coatings was also calculated and the dependence of CR on the weight percent of TiO₂ particles in the Cu-TiO₂ nanocomposite coating is shown in Fig. 9. It revealed that the Cu-TiO₂ nanocomposite coatings (CR=0.08 mm/year) were more corrosion resistant than electrodeposited copper coating (CR=1.175 mm/year) in 3.5% NaCl solution.

**Fig. 9** Dependence of CR on the weight percentage of TiO₂ particles in the Cu-TiO₂ nanocomposite coatings

Conclusions

The amount of TiO₂ nanoparticles in Cu-TiO₂ nanocomposite coatings increased with the addition of TiO₂ in the plating bath. The average crystallite size calculated was ~32 nm for electrodeposited copper and ~33 nm for Cu-TiO₂ nanocomposite coatings. The structure was crystalline fcc for both electrodeposited copper and Cu-TiO₂ nanocomposite coatings. The microhardness and wear resistance of the Cu-TiO₂ nanocomposite coatings were higher than electrodeposited copper. The charge transfer resistance (R_{ct}) of the Cu-TiO₂ nanocomposite coating was 1.15×10^3 ohm/cm², which was higher than that of electrodeposited copper. The Tafel polarization studies showed decreased corrosion currents for all the Cu-TiO₂ nanocomposite coatings than electrodeposited copper coating. The CR of Cu-TiO₂ nanocomposite coatings was less than that of electrodeposited copper coating. It revealed that Cu-TiO₂ nanocomposites were more corrosion resistant than electrodeposited copper in 3.5% NaCl solution.

References

1. Levin BF, Dupont JN, Marder AR (2000) *Wear Mater* 238:160. doi:10.1016/S0043-1648(99)00363-4
2. Hou KH, Ger MD, Wang LM, Ke ST (2002) *Wear* 253:994. doi:10.1016/S0043-1648(02)00222-3
3. Wu G, Li N, Zhou D, Mitsuo K (2004) *Surf Coat Tech* 176:157. doi:10.1016/S0257-8972(03)00739-4
4. Buelens C, Celis JP, Roos JR (1983) *J Appl Electrochem* 13:541. doi:10.1007/BF00617528
5. Greco VP (1989) *Plat Surf Finish* 76:68
6. Lee CC, Wan CC (1988) *J Electrochem Soc* 135:1930. doi:10.1149/1.2096182
7. Chen ES, Lakshmin GR, Sautter FK (1971) *Metall Trans* 2:937. doi:10.1007/BF02664222
8. Groza JR, Gibeling JC (1993) *Mater Sci Eng A* 171:115. doi:10.1016/0921-5093(93)90398-X
9. Stojak JL, Talbot JB (2001) *J Appl Electrochem* 31:559. doi:10.1023/A:1017558430864
10. Li J, Jing J, He H, Sun Y (2002) *J Mater Sci Lett* 21:939. doi:10.1023/A:1016073606681
11. Fujishima A, Rao TN, Tryk DA (2000) *J Photochem Photobiol C. Phytochem Rev* 1:1
12. Ito S, Deguchi T, Imai K, Iwasaki M, Tada H (1999) *Electrochem Solid-State Lett* 2:440. doi:10.1149/1.1390864
13. de Tacconi NR, Boyles AA, Rajeshwar K (2000) *Langmuir* 16:5665. doi:10.1021/la000037x
14. Deguchi T, Imai K, Matsui H, Iwasaki M, Tada H, Ito S (2001) *J Mater Sci* 36:4723. doi:10.1023/A:1017927021397
15. Thiemiig D, Bund A (2008) *Surf Coat Tech* 202:2976. doi:10.1016/j.surfcoat.2007.10.035
16. Li J, Sun Y, Sun X, Qiao J (2005) *Surf Coat Tech* 192:331. doi:10.1016/j.surfcoat.2004.04.082
17. Praveen BM, Venkatesha TV (2008) *Appl Surf Sci* 254:2418. doi:10.1016/j.apsusc.2007.09.047
18. Gomes A, Silva Pereira MI, Mendon MH, Costa FM (2005) *J Solid State Electrochem* 9:190. doi:10.1007/s10008-004-0573-2
19. Ramesh CS, Noor Ahmed R, Mujeebu MA, Abdullah MZ (2008) *Mater Des* (in press)
20. Wan YZ, Wang YL, Tao HM, Cheng GX, Dong XH (1998) *J Mater Sci Lett* 17:1251. doi:10.1023/A:1006539004598
21. Zhu JH, Liu L, Hu GH, Shen B, Hu WB, Ding WJ (2004) *Mater Lett* 58:1634. doi:10.1016/j.matlet.2003.08.040
22. Stankovic VD, Gojo M (1996) *Surf Coat Tech* 81:225. doi:10.1016/0257-8972(95)02486-7
23. Terzieva V, Fransaer J, Celis JP (2000) *J Electrochem Soc* 147:198. doi:10.1149/1.1393174
24. Ghorbani M, Mazaheri M, Khangholi K, Kharazi Y (2001) *Surf Coat Tech* 148:71. doi:10.1016/S0257-8972(01)01322-6
25. Celis JP, Roos JR (1977) *J Electrochem Soc* 124:1508. doi:10.1149/1.2133102
26. Fawzy MH, Ashour MM, Abd El-Halim AM (1995) *Trans Inst Metab Finish* 73:132
27. Bhalla V, Ramasamy C, Singh N, Pushpavanam M (1995) *Plat Surf Finish* 82:58
28. Celis JP, Roos JR, Buelens C (1987) *J Electrochem Soc* 134:1402. doi:10.1149/1.2100680
29. Fransaer J, Celis JP, Roos JR (1992) *J Electrochem Soc* 139:413. doi:10.1149/1.2069233
30. Lozano-Morales A, Podlaha EJ (2004) *J Electrochem Soc* 151: C478. doi:10.1149/1.1752934
31. Benea L, Mitoseriu O, Galland J, Wenger F, Ponthiaux P (2000) *Mater Corros* 51:491. doi:10.1002/1521-4176(200007)51:7<491::AID-MACO491>3.0.CO;2-C
32. Yang YL, Wang YD, Ren Y, He CS, Deng JN, Nan J, Chen JG, Zuo L (2008) *Mater Lett* 62:47. doi:10.1016/j.matlet.2007.04.086
33. Guo D, Zhang M, Jin Z, Kang R (2006) *J Mater Sci Technol* 22:514. doi:10.1179/174328406X100680
34. Singh VB, Pandey P (2005) *J New Mater Electrochem Syst* 8:299
35. Aruna ST, William Grips VK, Rajam KS (2009) *J Alloy Comp* 468:546. doi:10.1016/j.jallcom.2008.01.058
36. Guglielmi N (1972) *J Electrochem Soc* 119:1009. doi:10.1149/1.2404383
37. Szczygiel B, Kotodziej M (2005) *Trans Inst Metab Finish* 83:181. doi:10.1179/002029605X61658
38. Swanson HE, Tatge E (1953) *Natl Bur Stand Circulation* 1(15):539